

Article

Construction of Bronze $\text{TiO}_2/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 Ternary Composite Photocatalyst toward High Photocatalytic Performance

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Abstract: Research has demonstrated that the formation of composites of titanium dioxide (TiO_2) with silver phosphate (Ag_3PO_4) through the construction of heterojunctions can expand its light absorption range and suppress the recombination of photogenerated electron–hole pairs, thereby improving the photocatalytic performance. However, this method offers only limited performance improvements, and the composite photocatalysts are costly due to the expensive Ag_3PO_4 . In this study, Ti_3C_2 MXene, which has good hydrophilicity and excellent electrical conductivity, is first used to form Schottky junction composites with bronze TiO_2 ($\text{TiO}_2(\text{B})$) via electrostatic self-assembly. Then, Ag_3PO_4 quantum dots were further formed on the surface of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene by in situ self-growth, and Ag_3PO_4 formed heterojunctions and Schottky junctions with $\text{TiO}_2(\text{B})$ and Ti_3C_2 MXene, respectively. Finally, a ternary composite photocatalyst $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 was jointly constructed by these functional junctions. Under the synergistic effect of these functional junctions, the mobility and fast separation performance of photogenerated electron–hole pairs of the composite photocatalyst were significantly improved, the recombination of photogenerated electron–hole pairs was effectively suppressed, and the light absorption performance was enhanced. As a result, the composite photocatalyst exhibited excellent photocatalytic performances.

Keywords: ternary composite photocatalyst; heterojunction; photocatalytic degradation; bronze TiO_2 ; Ti_3C_2 MXene



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1. Introduction

The titanium dioxide (TiO_2) photocatalyst has been widely studied as a safe, inexpensive, and environmentally friendly photocatalyst. However, it does not exhibit high photocatalytic performances since it is only active for ultraviolet (UV) light due to its wide forbidden band gap and because the photogenerated electron–hole pairs easily recombine [1–4]. Previous studies have demonstrated that the compositing of TiO_2 and silver phosphate (Ag_3PO_4) via the construction of heterojunctions can expand its light absorption range and suppress the recombination of photogenerated electron–hole pairs, thereby improving the photocatalytic performance. However, such a performance improvement is limited, and the composite catalysts are costly due to the expensive Ag_3PO_4 [5–8]. Ti_3C_2 MXene is a new two-dimensional material with good hydrophilicity and excellent electrical conductivity. It can be used as a co-catalyst with other semiconductors to form Schottky junctions, which can not only effectively expand the light absorption range but also

promote the rapid separation of photogenerated electron-hole pairs, thus enhancing the photocatalytic performances of the composite photocatalysts [9–12].

To obtain high-performance and low-cost photocatalysts, $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ternary composite photocatalysts were constructed in this study. First, a $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene composite was constructed by electrostatic self-assembly of strip-like $\text{TiO}_2(\text{B})$ and lamellar Ti_3C_2 MXene, and then Ag_3PO_4 quantum dots were formed on the surface of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene by in situ growth, thereby forming the $\text{TiO}_2/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ternary composite photocatalyst. The composite photocatalyst exhibited excellent photocatalytic performances, which first increased and then decreased with the increase in the content of the co-catalyst Ti_3C_2 MXene. The composite photocatalyst reached the best performance when the content of Ti_3C_2 MXene was 20% of the mass of $\text{TiO}_2(\text{B})$. Compared with the common TiO_2 phases (anatase, brookite, and rutile), bronze TiO_2 has a good electrical conductivity and a loose, porous structure, which facilitates the rapid migration of carriers and the formation of reaction sites in heterojunction photocatalysts, ultimately enhancing the performances of composite photocatalysts. Therefore, bronze TiO_2 was used in this study as a component of the composite photocatalyst [13–16].

2. Results and Discussion

As shown in Figure 1a, the standard PDF cards of Ag_3PO_4 , $\text{TiO}_2(\text{B})$, and Ti_3AlC_2 were JCPDS#06-0505, JCPDS#35-0088, and JCPDS#015-3266, respectively. The diffraction pattern of the Ti_3C_2 MXene demonstrated that, after HF etching, the strongest diffraction peak of Ti_3AlC_2 was at $2\theta = 28.99^\circ$, corresponding to the (104) crystal plane disappearing, while the diffraction peaks at $2\theta = 9.52^\circ$ and 19.12° corresponding to the (002) and (004) crystal planes, respectively, were enhanced, and their positions shifted towards smaller diffraction angles. Referring to the literature [17–20], the characteristics of the above diffraction peaks indicated that the Al atomic layer of Ti_3AlC_2 was successfully removed via etching to obtain the Ti_3C_2 MXene in this study. A comparison of the XRD (X-ray diffraction) patterns of the $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$, Ti_3C_2 MXene, and $\text{TiO}_2(\text{B})$ revealed that the diffraction peaks of the $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ spectrum were mainly those of $\text{TiO}_2(\text{B})$, which appeared weakly and were only observed at $2\theta = 9.14^\circ$, corresponding to the (002) crystal plane of the Ti_3C_2 MXene. In addition, compared with that of the $\text{TiO}_2(\text{B})$, the half-peak width of the diffraction peak of the $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ spectrum increased, which may have been due to the low content of the Ti_3C_2 MXene and the relatively weak diffraction peaks. Figure 1b shows that the diffraction pattern of the ternary composite was mainly composed of diffraction peaks corresponding to Ag_3PO_4 , while those corresponding to the Ti_3C_2 MXene and $\text{TiO}_2(\text{B})$ were not observed, which was mainly because Ag_3PO_4 had good crystallinity and exhibited strong diffraction compared to the Ti_3C_2 MXene and $\text{TiO}_2(\text{B})$.

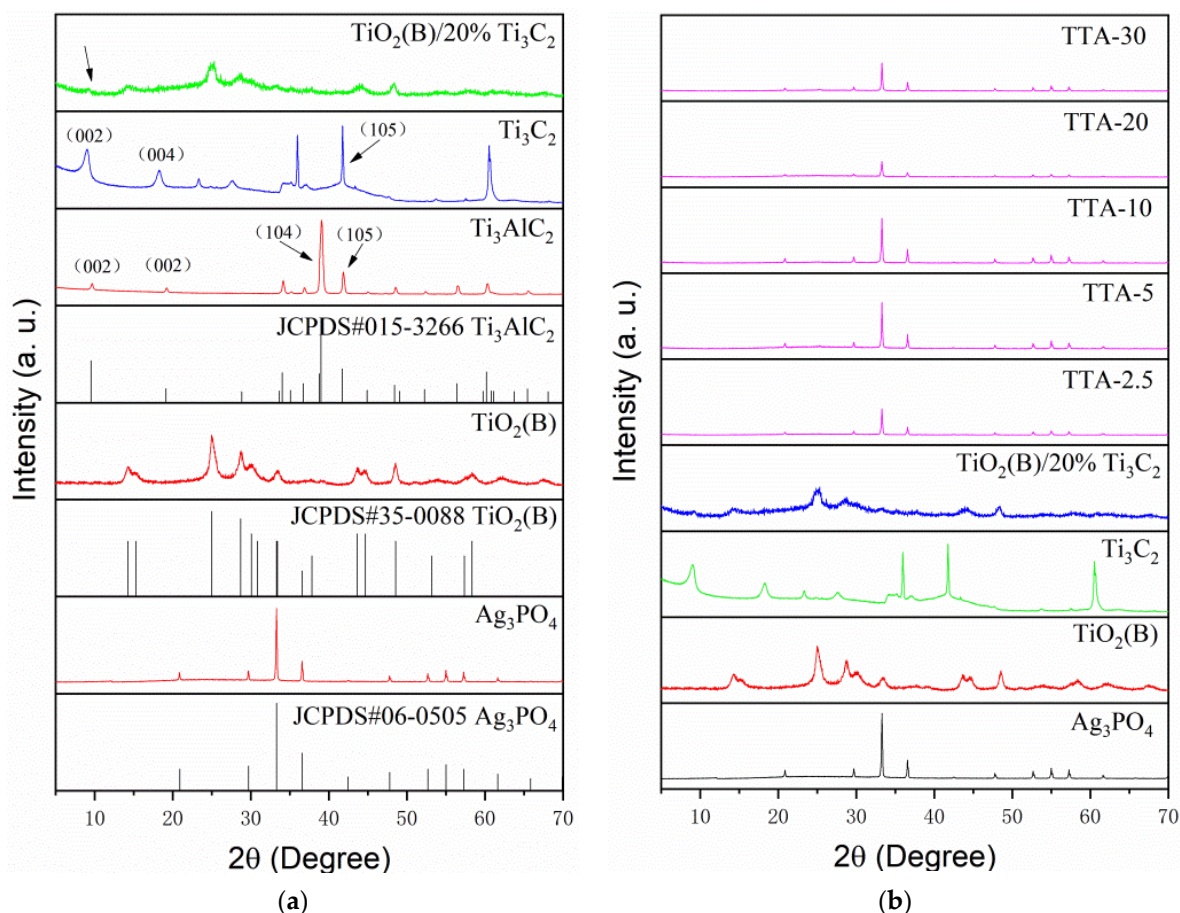


Figure 1. (a) XRD diffraction patterns of Ag_3PO_4 , $\text{TiO}_2(\text{B})$, Ti_3AlC_2 , Ti_3C_2 and $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$; (b) XRD diffraction patterns of TTA-2.5, TTA-5, TTA-10, TTA-20 and TTA-30.

Figure 2a shows the SEM (scanning electron microscope) image of the Ti_3C_2 MXene prepared from Ti_3AlC_2 . Since the Al atomic layer in the Ti_3AlC_2 was removed by etching, the Ti_3C_2 MXene had an accordion-like morphology, which is typical of Ti_3C_2 MXene. Figure 2b shows the SEM image of the experimentally prepared bronze TiO_2 , which had a strip-like morphology. Figure 2c shows the SEM image of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ternary composite, showing that the Ti_3C_2 MXene was closely combined with the $\text{TiO}_2(\text{B})$, with a large number of fine Ag_3PO_4 particles distributed on their surfaces, which are shown more clearly in the TEM (Transmission Electron Microscope) images in Figure 2d,f. Figure 2e,g show the HRTEM (High Resolution Transmission Electron Microscope) images, from which we can observe the lattices of Ag_3PO_4 (211), bronze TiO_2 (110), and Ti_3C_2 (105) as well as heterojunctions formed in the overlapping regions of these lattices. Figure 2h shows the mapping diagram of the elements in the composite, revealing that the composite contained five elements (Ti, O, Ag, P, and C), the distributions of which were consistent with the above phase and morphological analysis.

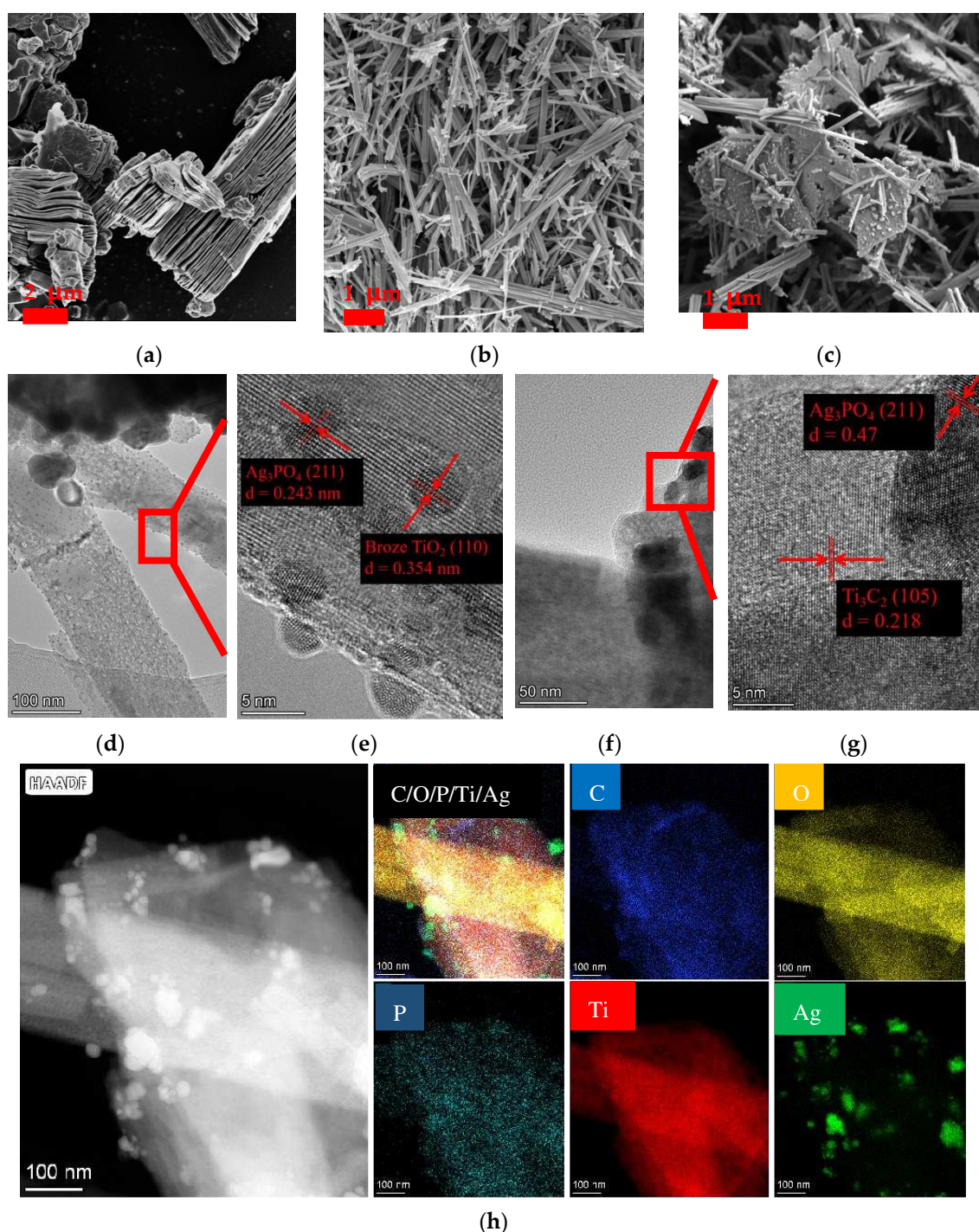


Figure 2. (a) SEM of Ti_3C_2 MXene; (b) SEM of $\text{TiO}_2(\text{B})$; (c) SEM of $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ; (d,f) TEM of $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ; (e,g) HRTEM of $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ; (h) mapping and HAADF (high angle annular dark field image) of samples.

Figure 3a shows the degradation curves of the samples. The photocatalytic performance of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 composite was significantly higher than those of the $\text{TiO}_2(\text{B})$, Ag_3PO_4 and $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene and is significantly higher than that of photocatalyst P_{25} sold in the market. The photocatalytic performance of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 composite first increased with the increase in the content of the Ti_3C_2 MXene and reached the highest value when the content was increased to 20%, beyond which further increases and lowered the photocatalytic performance of the composite instead. Figure 3b shows a linear fit of the degradation kinetics of the sam-

ples. According to the Langmuir–Hinshelwood model, the degradation reaction can be considered to be a first-order reaction at low RhB concentrations, and thus, the fitting of the degradation kinetics can be simplified to a linear case. The degradation reaction in this experiment satisfied the apparent first-order reaction rate equation $\ln(C/C_0) = -k \cdot t$, where k is the apparent first-order reaction rate constant. Thus, $\ln(C/C_0)$ is a function of the irradiation time t . The k values of the $\text{TiO}_2(\text{B})$, $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene, Ag_3PO_4 and P_{25} were low, with values of 0.008, 0.011, 0.085 and 0.04 min^{-1} , respectively, while the k value corresponding to $\text{TTA}(\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2 \text{ MXene}/\text{Ag}_3\text{PO}_4)$ was significantly higher, which gradually increased from $k_{\text{TTA-2.5}} = 0.182 \text{ min}^{-1}$ to the maximum $k_{\text{TTA-20}} = 0.345 \text{ min}^{-1}$ and then decreased to $k_{\text{TTA-30}} = 0.241 \text{ min}^{-1}$.

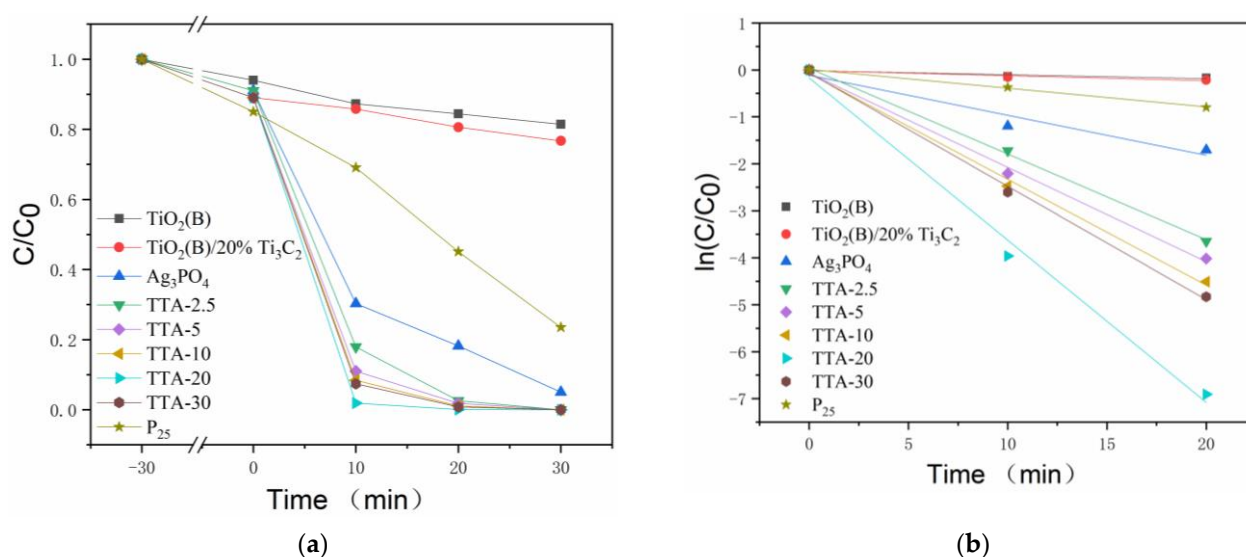


Figure 3. (a) Photocatalytic degradation curves; (b) linear fittings of degrading kinetics under different catalytic conditions.

Figure 4 shows the UV-Vis absorption spectra of the samples. Compared with $\text{TiO}_2(\text{B})$, the visible light absorption capacity was clearly improved after forming a composite of $\text{TiO}_2(\text{B})$ and Ti_3C_2 MXene, and the light absorption performance of sample TTA-20 obtained by further incorporating Ag_3PO_4 was even higher. The band gaps of the Ti_3C_2 , $\text{TiO}_2(\text{B})$, $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene, and TTA-20 were calculated to be 1.67, 3.14, 3.23, and 3.32 eV, respectively, based on the Tauc plot. These results confirmed the band gap reduction by the formation of the composite, which is an important reason for the improved light absorption performances of the composite materials.

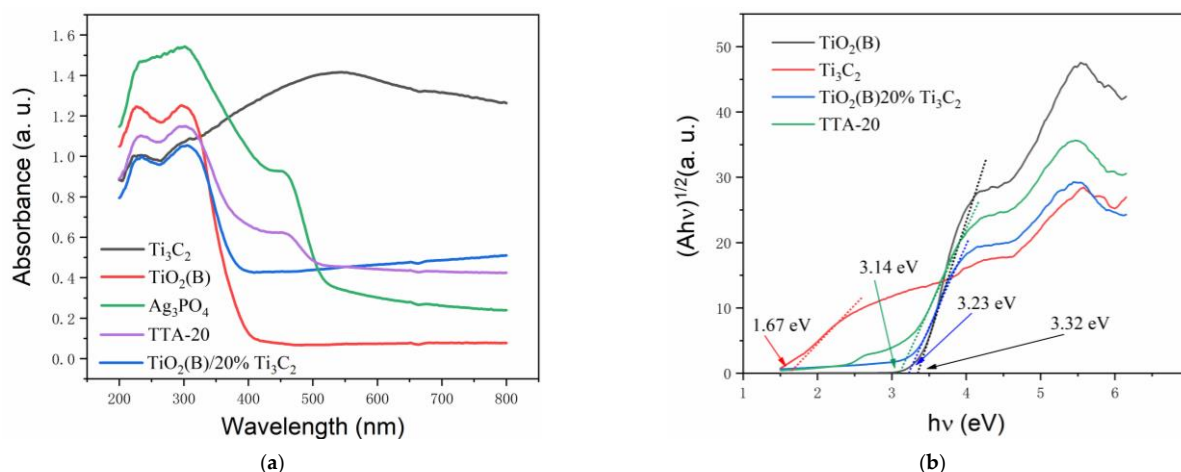


Figure 4. (a) Ultraviolet-visible (UV-Vis) absorption spectra and (b) band gaps of samples.

As shown in Figure 5, the fluorescence emission spectra as well as the transient photocurrent response and electrochemical impedance Nyquist plots of the samples were obtained to analyze the influence of the formation of composites on the recombination and migration of photogenerated electron–hole pairs in the samples. Figure 5a shows that the fluorescence of the $\text{TiO}_2(\text{B})/20\% \text{Ti}_3\text{C}_2$ MXene was significantly weaker than that of the $\text{TiO}_2(\text{B})$, and the fluorescence of TTA-20 ($\text{TiO}_2(\text{B})/20\% \text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4) was the weakest. This was because the highly conductive Ti_3C_2 MXene was composited with $\text{TiO}_2(\text{B})$ to form Schottky junctions on the contact surface, which caused the fast migration of photogenerated electrons to the Ti_3C_2 MXene, achieving rapid separation of photogenerated electron–hole pairs and suppressing the recombination. By forming the Ti_3C_2 MXene and $\text{TiO}_2(\text{B})$ composite, Ag_3PO_4 quantum dots were introduced on the surfaces of the Ti_3C_2 MXene and $\text{TiO}_2(\text{B})$ by in situ self-growth, the Ag_3PO_4 and Ti_3C_2 MXene formed Schottky junctions, and the Ag_3PO_4 and $\text{TiO}_2(\text{B})$ formed heterojunctions. These junctions created a synergistic effect with the Schottky junctions formed by the Ti_3C_2 MXene and $\text{TiO}_2(\text{B})$, further promoting the rapid migration and separation of photogenerated electrons and greatly suppressing the recombination of photogenerated electron–hole pairs. Figure 5b shows the transient photocurrent responses of the samples. The photocurrents of the samples were enhanced by compositing. The photocurrent of TTA-20 was the strongest, followed by those of the $\text{TiO}_2(\text{B})/20\% \text{Ti}_3\text{C}_2$ MXene, Ag_3PO_4 , and $\text{TiO}_2(\text{B})$, in that order. The photocurrent enhancement pattern was consistent with the fluorescence weakening pattern, and the photocurrent enhancement also indicated that the recombination of photogenerated electron–hole pairs was effectively suppressed. Thus, Figure 5b further confirms the results and analysis of Figure 5a. Figure 5c shows the Nyquist plot of the electrochemical impedances of the samples. In general, the smaller the curvature radius of the curve is, the greater the electron mobility of the sample is. The mobility was significantly enhanced by forming the composite. The electron mobility of TTA-20 was the largest, followed by that of the $\text{TiO}_2(\text{B})/20\% \text{Ti}_3\text{C}_2$ MXene, and both were larger than those of the $\text{TiO}_2(\text{B})$ and Ag_3PO_4 .

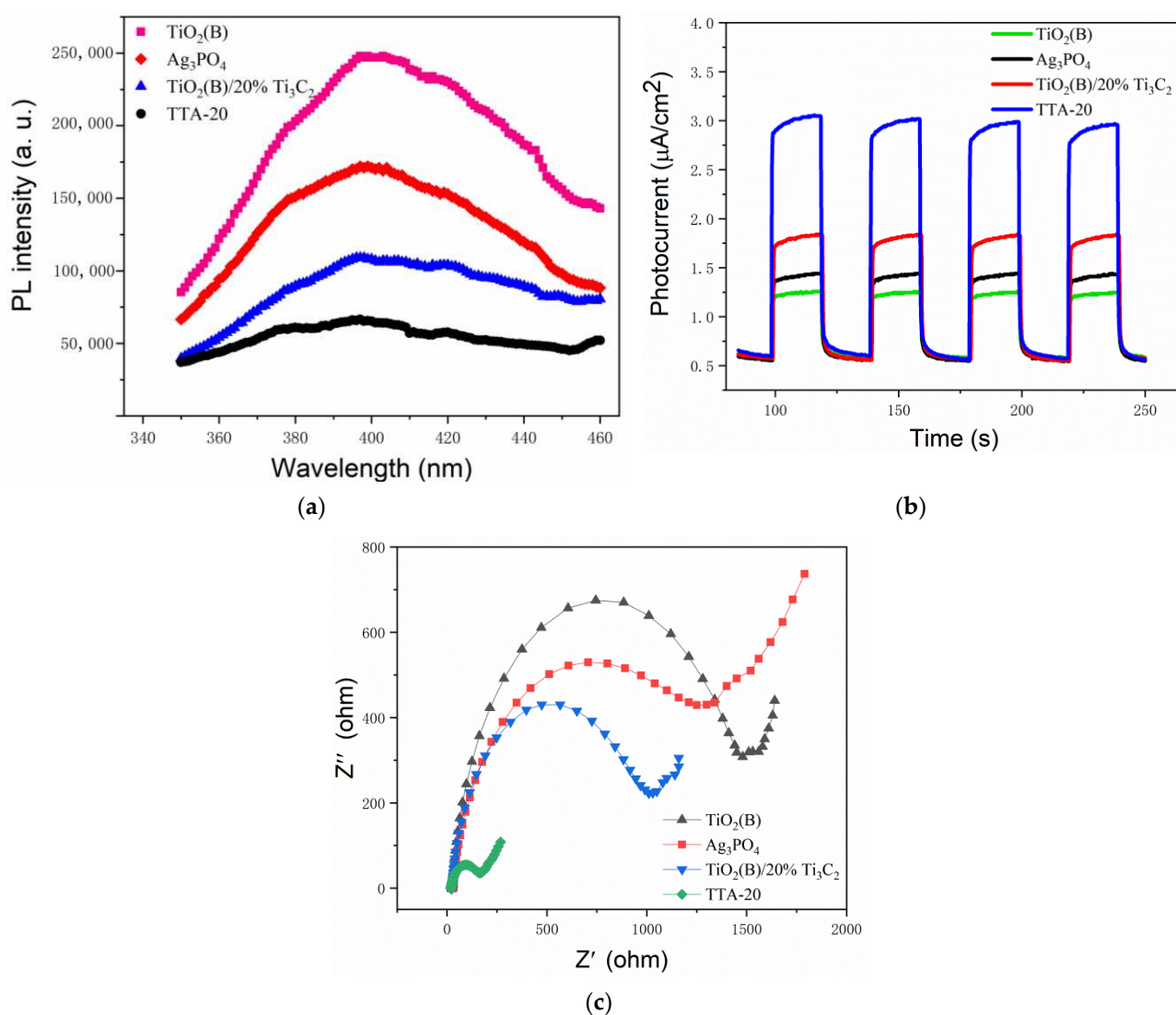


Figure 5. (a) Photoluminescence (PL) spectra; (b) transient photocurrent responses; (c) electrochemical impedance Nyquist.

3. Materials and Experiments

3.1. Preparation of Materials and Samples

The following reagents were used: Rhodamine B (RhB, $\text{C}_{28}\text{H}_{31}\text{ClN}_2\text{O}_3$) (Solarbio, Beijing, China); titanium aluminum carbide (Ti_3AlC_2) (Xianfeng Nanomaterials, Nanjing, China); disodium hydrogen phosphate 12-water ($\text{Na}_2\text{HPO}_4 \cdot 12\text{H}_2\text{O}$, Jinshan Chemical Reagent, Chengdu, China); silver nitrate (AgNO_3) (Sinopharm Chemical Reagent, Shanghai, China); and sodium hydroxide (NaOH), hydrofluoric acid (HF), and nano- TiO_2 (Aladdin Industrial, Shanghai, China). The deionized (DI) water used for the experiments was prepared in our laboratory. All reagents were of analytical grade and were used without further purification.

Ti_3C_2 MXene was prepared from Ti_3AlC_2 by HF etching as follows. An appropriate amount of 40% (mass fraction) Ti_3AlC_2 was slowly added to the HF acid solution at a ratio of 1 g/20 mL and stirred at room temperature for 24 h. The prepared product was separated by centrifugation at 4000 rpm and rinsed repeatedly with DI water to near neutrality.

The preparation of $\text{TiO}_2(\text{B})$ was as follows. A 10 M NaOH aqueous solution was prepared. Then, raw TiO_2 was added, stirred well, poured into a hydrothermal reactor, and heated up to 200 °C in a blast drying oven and held for 24 h. Next, the reaction product was removed, rinsed with a large amount of DI water to near neutrality, soaked with 0.1 M

dilute hydrochloric acid for 72 h, rinsed again with a large amount of DI water to near neutrality, dried at 60 °C for 10 h, and finally calcined in a muffle furnace at 500 °C for 2 h to obtain $\text{TiO}_2(\text{B})$.

The $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ternary composite was prepared as follows. About 100 mg of $\text{TiO}_2(\text{B})$ and 20 mg of Ti_3C_2 MXene were added to DI water and sonicated for 4 h to allow the formation of a composite of $\text{TiO}_2(\text{B})$ and Ti_3C_2 MXene by electrostatic self-assembly, which was followed by drying in a vacuum drying oven to obtain the $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene composite. $\text{Na}_2\text{HPO}_4 \cdot 12\text{H}_2\text{O}$ and $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene were added to DI water at a molar ratio of 0.8:1 (note that the number of moles of $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene was the sum of the numbers of moles of the two components) and mixed well. A solution was prepared using AgNO_3 in a molar ratio of 3:1 with $\text{Na}_2\text{HPO}_4 \cdot 12\text{H}_2\text{O}$, and then the solution was slowly added dropwise using a syringe to the mixture of $\text{Na}_2\text{HPO}_4 \cdot 12\text{H}_2\text{O}$ and $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene. The mixture was stirred continuously to form Ag_3PO_4 quantum dots on the surface of the $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene via in situ self-growth. The product was rinsed with DI water and dried in the vacuum drying oven to obtain $\text{TiO}_2(\text{B})/20\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 , which was labeled as TTA-20. The preparation flow chart is shown in Figure 6. Using the same method, $\text{TiO}_2(\text{B})/2.5\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 , $\text{TiO}_2(\text{B})/5\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 , $\text{TiO}_2(\text{B})/10\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 , and $\text{TiO}_2(\text{B})/30\%\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 were prepared and labeled as TTA-2.5, TTA-5, TTA-10, and TTA-30, respectively.

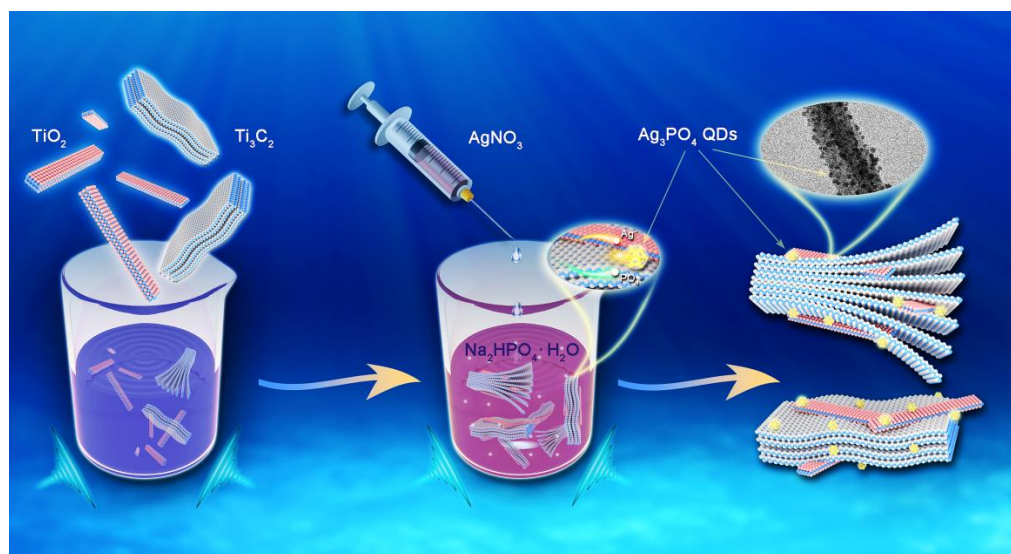


Figure 6. Preparation flow chart of $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 ternary composite.

3.2. Analytical Tests

An X-ray powder diffraction (XRD, BRUCKER D8 ADVANCE, Bruker, Bremen, Germany) was used to test and analyze the phases of the samples using the following test parameters: Cu target, wide-angle diffraction, scan range of 5°–70°, scan rate of 10°/min, test wavelength of 1.5406 Å, tube voltage of 40 kV, and tube current of 40 mA. Field-emission scanning electron microscopy (FE-SEM, Gemini SEM 300, Zeiss, Oberkochen, Germany) was used to characterize the morphologies and structures of the samples at an operating voltage of 3 kV. Field-emission transmission electron microscopy (FEI Talos F200X, FEI, Hillsboro, OR, USA), with a maximum magnification of 1.1 million times and a limit resolution of 0.12 nm, was used to observe and analyze the morphologies and structures of the samples at an operating voltage of 200 kV. An ultraviolet–visible (UV–Vis) spectrophotometer (UV-3600, Shimadzu company, Kyoto, Japan), with a test range of 200–2500 nm, was used to test the photocatalytic degradation of RhB and obtain the UV–Vis absorption spectra of the samples. An electrochemical workstation (CHI-760E, Shanghai

Chenhua Instrument, Shanghai, China) was used to test the photocurrents and impedances of the samples.

Two liters of 20 mg/L RhB solution were prepared, and 100 mL was taken for each experiment, added to 20 mg of the sample, and sonicated for 30 min. Then, this mixture was placed under a light source with an intensity of 600 W/m^2 (at the liquid surface) in a solar simulator (Solar-500Q, Newbit, Beijing, China) to start the photocatalytic degradation experiment, which lasted for 60 min. During the experiment, 5 mL of liquid was taken from the mixture every 10 min and subjected to centrifugation at 10,000 rpm for 10 min. After centrifugation, the supernatant was taken and its absorbance A_n was measured by the UV-Vis spectrophotometer. There was a direct proportionality relation between the solution concentration C_n and A_n , $C_n/C_0 = A_n/A_0$, where C_0 was the concentration of 20 mg/L of the RhB solution, and A_0 was the absorbance of the 20 mg/L RhB solution.

4. Conclusions

In this study, the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene Schottky junction composite was first prepared by electrostatic self-assembly, and then Ag_3PO_4 quantum dots were generated on the surface of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene by in situ self-growth, thereby forming $\text{Ag}_3\text{PO}_4/\text{TiO}_2(\text{B})$ heterojunctions and $\text{Ag}_3\text{PO}_4/\text{Ti}_3\text{C}_2$ MXene Schottky junctions. Through the successful construction and synergy of these functional junctions (As shown in Figure 7), the light absorption capacity of the composite photocatalyst was greatly enhanced, and the recombination of photogenerated electron–hole pairs was effectively suppressed, thus substantially improving the photocatalytic degradation performance of the composite photocatalyst. It was further found that the photocatalytic performance of the $\text{TiO}_2(\text{B})/\text{Ti}_3\text{C}_2$ MXene/ Ag_3PO_4 composite increased with the increase in the Ti_3C_2 MXene content, and it reached the maximum when the Ti_3C_2 MXene content increased to 20%, beyond which the photocatalytic performance of the composite decreased as the Ti_3C_2 MXene content continued to increase.

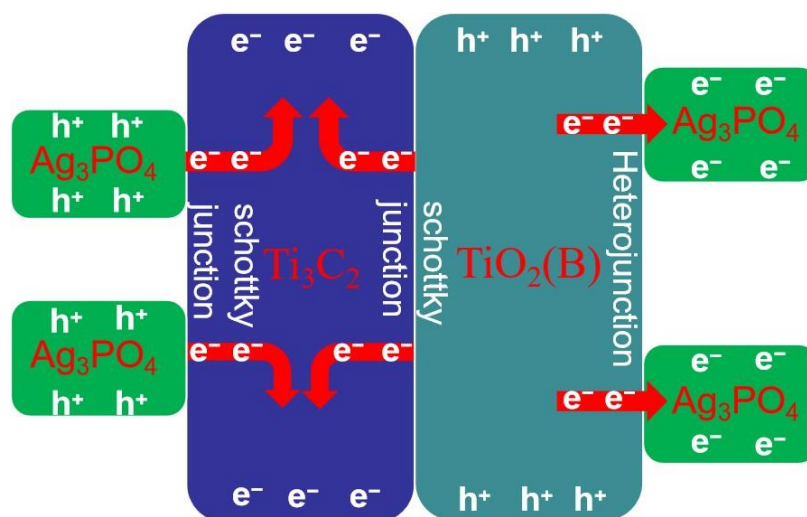


Figure 7. Mechanism of enhanced photocatalytic properties.

Author Contributions: Conceptualization, methodology, S.W. and Y.C.; software, Y.L. (Yong Li); validation, Y.L. (Yong Li) and Y.L. (Yanfang Liu); formal analysis, Y.L. (Yong Li) and M.Z.; investigation, Y.L. (Yong Li), Y.L. (Yanfang Liu) and M.Z.; resources, S.W. and Y.C.; data curation, S.W. writing—original draft preparation, Y.L. (Yong Li), M.Z. and X.L.; writing—review and editing, Y.L. (Yong Li), M.Z. and Y.L. (Yanfang Liu); visualization, Q.Z. (Qinghua Zhao) and Q.Z. (Qianyu Zhou); project administration, Y.L. (Yong Li); funding acquisition, Y.L. (Yong Li) and S.W. All authors have read and agreed to the published version of the manuscript.

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Data Availability Statement: The data presented in this study are available on request from the corresponding author.

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